

## Characteristics of $\alpha$ Precipitated Phase and Tensile Properties of Cast Ti5553 Alloy

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**Abstract:** Ti5553 (Ti-5Al-5V-5Mo-3Cr) not only has high strength, but also has good mould filling fluidity, which is a typical high strength casting titanium alloy. The mechanical properties and deformation behavior of Ti5553 alloy mainly depend on the coupling effect of  $\alpha$  phase characteristics and residual  $\beta$  matrix. The current research mostly focuses on the deformed alloy and the dependence of the mechanical properties of cast Ti5553 alloy on  $\alpha$  phase characteristics is still unclear. In this paper, the room temperature tensile deformation behaviour of cast Ti5553 alloy and its relationship with  $\alpha$  precipitated phase characteristics under different heat treatment process are investigated. The results show that: the HCP+FCC structure within the  $\alpha$  lamellae of the alloy in the hot isostatic state activates numerous slip systems and the coordinated deformation of  $\alpha$  twins and FCC phases leads to yield strength, tensile strength, and elongation of 955 MPa, 1020 MPa, and 10%, respectively; After solid solution aging, a large number of fine acicular  $\alpha$  phases precipitate on the  $\beta$  matrix, and the alloy has high strength but very low ductility; the double annealed state alloy has the best strength-ductility match, and the yield strength, tensile strength, and elongation reach 1020 MPa, 1080 MPa, and 8%, respectively. The multiscale  $\alpha$  lamellae intralaminar slip shear and  $\alpha$  twins are the structural roots of their good strength-ductility.

**Keywords:** Cast Ti5553 alloy,  $\alpha$  phase characteristics, Tensile properties, Deformation behavior

### 1 Introduction

Metastable  $\beta$  titanium alloys have great potential for application in fields such as aerospace and biomedicine due to their high specific strength, good fatigue resistance, and excellent processing performance<sup>[1,2]</sup>. Ti5553 alloy is expected to become a typical high-strength cast titanium alloy. The mechanical properties and deformation behavior of Ti5533 alloy actually depends on the characteristics of the  $\alpha$  precipitated phase and the coupling of residual  $\beta$  phase. They are highly sensitive to heat treatment processes. In recent years, the strength-ductility of metastable  $\beta$  alloys is mostly improved by heat treatment to obtain the multiscale  $\alpha$  phase<sup>[3]</sup>. Currently, most of the related studies focus on deformed alloys, and the dependence of the mechanical properties of cast Ti5553 alloy on the characteristics of  $\alpha$  precipitated phases is not clear.

Therefore, in this paper, the room temperature tensile deformation behaviour of cast Ti5553 alloy under different heat treatment process and its relationship with  $\alpha$  precipitated phase characteristics are investigated.

### 2 Experimental procedure

The cast Ti5553(Ti-5.27Al-5.15V-5.25Mo-3.16Cr-0.22O) alloy used in this study, was prepared through investment casting. To address the hole defects in the casting process, a hot isostatic pressing treatment (HIP,820 °C/130 MPa/2 h) was applied to the Ti5553 alloy cast ingot. Three kinds of heat treatment processes, annealing (700 °C/2 h/AC), double annealing (865 °C/30min→600 °C/4 h/AC) and solid solution ageing (880 °C/30 min/AC+600 °C/4 h/AC), were carried out on the HIP-Ti5553 alloy.

### 3 Result and discussion

#### Microstructure and mechanical properties

The average thickness of the  $\alpha$  lamellae in the HIP-Ti5553 alloy is  $0.74\pm 0.20\ \mu\text{m}$ , and there is no fine acicular  $\alpha$  phase on the  $\beta$  matrix. The  $\alpha$  lamellae within the  $\beta$  matrix are slightly coarsened after annealing. After double annealing,  $\beta$  matrix precipitated lamellar  $\alpha$  with “fibrous root morphology”<sup>[4]</sup> and fine acicular  $\alpha$  phases. The average thickness of the  $\alpha$  lamellae reaches  $1.16\pm 0.20\ \mu\text{m}$ , and the thickness of the fine acicular  $\alpha$ -phase is about 150~250 nm, and the length is 0.75~1.25  $\mu\text{m}$ . After solid solution ageing, a large number of V-shaped or triangular fine acicular  $\alpha_s$  phases precipitated in the  $\beta$  matrix. The average thickness of the fine acicular  $\alpha_s$  is below 100 nm, and the length is about 0.1-0.3  $\mu\text{m}$ . The mechanical properties of the alloys in different states are shown in Fig.1. Annealed alloys have the highest elongation but low strength; the alloys after solid solution ageing have the highest strength but almost brittle fracture; the alloys in the double-annealed state had the best strength-ductility match, with yield strength, tensile strength and elongation up to 1020 MPa, 1080 MPa and 8%, respectively.

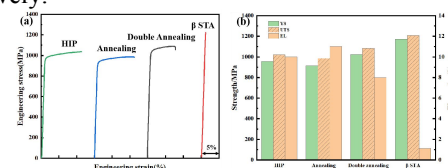


Fig. 1 Comparison of tensile stress-strain curves (a) and properties (b) for cast Ti5553 alloy after different

heat treatment process

#### 4 Deformation behaviour

In order to further analyze the reasons for the good matching of strength-ductility of the HIP-Ti5553 and double annealed Ti5553, the near-fracture microstructure was analyzed by TEM. Fig.2 shows the microstructural characterization of the HIP-Ti5553 with different strains. Due to the high temperature and high pressure environment provided by the hot isostatic pressure, there are a small number of  $\alpha$  twins and FCC phases in the  $\alpha$  lamellae of the HIP-Ti5553 before tensile deformation. A large number of dislocations and slip bands can be seen in both the  $\alpha$  phase and the  $\beta$  matrix when the tensile strain is 2%, in which the  $\alpha$  phase is surrounded by neatly arranged misfit dislocations. When the tensile strain is increased to 6%, not only a large number of dislocations are found within the  $\alpha$  phase of HIP-Ti5553 alloy, but also parallel distribution of  $\alpha$  twins are observed. When the alloy is stretched to fracture, tiny twins of interconnected and zigzag shapes appear within the  $\alpha$  phase, and there is a high density of stacking fault within some of the  $\alpha$  twins and the  $\alpha$  matrix between  $\alpha$  twins. The  $\alpha$  twins and stacking fault can form the Lomer-Cottrell lock<sup>[5]</sup>. The initially existing FCC phase deformed during tensile process, resulting in FCC twins. The HCP+FCC structure within the  $\alpha$  lamellae activates numerous slip systems and the coordinated deformation of  $\alpha$  twins and FCC phases resulting in an ideal strength- ductility match.

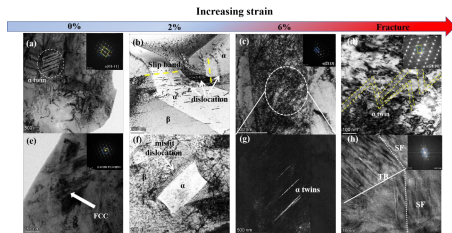


Fig. 2 TEM images of HIP-Ti5553 with different tensile strains:(a,e) 0%;(b,f) 2%;(c,g) 6%;(d,h) Fracture

Fig.3 shows the near-fracture microstructure of the double annealed alloy, which can be clearly seen that there are a small number of twins in the fine acicular  $\alpha$  phase, and the coarse  $\alpha$  lamellae has obvious slip shear bands and dislocations, which also verifies that the  $\alpha$  phase with the "fibrous root morphology"<sup>[4]</sup> has good deformation ability, and there are high-density dislocations near the fine acicular  $\alpha$ . The "fibrous root morphology"  $\alpha$  and a small number of  $\alpha$  twins ensure the ductility of the alloy, and the distribution

of fine acicular  $\alpha$  to dislocations in a vertical or parallel manner can better impede the alloy's strength improvement.

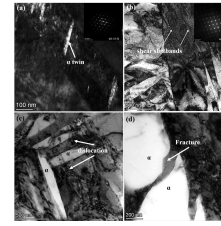


Fig. 3 TEM images of near fracture surface of cast Ti5553 alloy after double annealing: (a)  $\alpha$  twins; (b) shear slip bands (c) fine acicular dislocations near  $\alpha$  phase; (d) Fracture of  $\alpha$  phase

#### 5 Conclusion

After double annealing, the cast Ti5553 alloy has a best strength- ductility match by precipitating multi-scale  $\alpha$  phase and inducing  $\alpha$  twins with room-temperature tensile yield strength and elongation up to 1020 MPa and 8%.

#### 6 Acknowledgments

This project was financially supported by the National Natural Science Foundation of China (U21A2050), Shaanxi Provincial Innovation Capacity Support Plan (2023-CX-TD-4), the State Key Laboratory of Solidification Processing (SKLSP202201) and ND Basic Research Funds, NPU (G2022WD).

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